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Structural complexity in indium selenides prepared using bicyclic amines as structure-directing agents†

S. J. Ewinga and P. Vaqueiro*a,b

The synthesis and characterization of five new indium selenides, [C₆H₄N₂][In₃Se₁₅(Se₂)₃] (1–2), [C₆H₄N₂][C₆H₄N₂][In₁₀Se₁₅(Se₂)₃] (3), [C₇H₁₀N][C₆H₄N₂][N₃]In₉Se₁₄ (4) and [enH][NH₄][In₃Se₂] (5), are described. These materials were prepared under solvothermal conditions, using 1,8-diazabicyclo-[5.4.0]undec-7-ene (DBU) and 1,4-diazabicyclo[2.2.2]octane (DABCO) as structure-directing agents. Compounds 1–4 represent the first examples of ribbons in indium selenides, and 4 is the first example of incorporation of an alkali metal complex. Compounds 1, 2 and 4 contain closely related [In₅Se₈+4(Se₂)]₄⁻ ribbons which differ only in their content of (Se₂)²⁻ anions. These ribbons are interspersed by organic counterions in 1 and 2, while in 4 they are linked by highly unusual [Na(DABCO)]⁺ units into a three-dimensional framework. Compound 3 contains complex ribbons, with a long repeating sequence of ca. 36 Å, and 4 is a non-centrosymmetric three-dimensional framework, formed as a consequence of the decomposition of DABCO into ethylenediamine (en) and ammonia.

Introduction

Open-framework chalcogenides are of considerable interest due to their potential as multifunctional materials in which semiconductivity and porosity coexist, and have already shown promising behaviour as photocatalysts, fast-ion conductors and for the selective ion exchange of heavy metals. Moreover, the primary building units in many of these chalcogenides lack an inversion centre, and hence are ideal building blocks for non-centrosymmetric solids, which may find applications in second-order nonlinear optics. Significant progress has already been made in the synthesis of open-framework indium sulfides, which are frequently built from supertetrahedral clusters, but very few indium selenides contain supertetrahedral clusters as secondary building units. This includes three-dimensional frameworks with the double- and single-diamond topologies, and [NH₄][InSe₂], which is a non-centrosymmetric material with nonlinear optical behaviour. Vertex-linking of In₅Se₈⁻ tetrahedra can lead to other secondary building units, such as the non-nuclear indium clusters found in the microporous structure of [NH₄][In₁₂Se₂₀], or to the helical chains found in the [In₃₃Se₅₆]₁³⁻ open framework. Edge-sharing of In₅Se₈⁻ tetrahedra is also possible, as exemplified by [C₆H₁₂N][In₇Se₁₂], in which edge- and vertex-sharing tetrahedra coexist. When In₅Se₈⁻ tetrahedra are linked exclusively by their edges, In₅Se₈⁻ chains are obtained. These have been found in a polymeric form of [NH₄][InSe₂] and in [bdah][InSe₂] (where bdha = 1,4-butanediamine). Recently, it has been found that replacement of Se⁵⁻ with perselenide (Se₂)²⁻ anions in the [In₅Se₈⁻] chains can occur, resulting in the formation of chains in which four and five-membered rings alternate, as exemplified by [C₆H₁₂N][In₇Se₈(Se₂)], (M[phen])₃In₅Se₈H₂Oₙ (where M = Ni, Fe; phen = 1,10-phenanthroline) and (Mn[en]₃)[In₅Se₈].

Despite the ongoing research efforts on the solvothermal synthesis of new indium selenides, the number of reported phases has remained low. This may be related to the low selectivity with which a particular structure is crystallised with a specific structure-directing agent. For instance, we have found that frameworks such as [NH₄][In₁₃Se₂₀] or [In₃₃Se₅₆] can be obtained using a wide variety of structure-directing...
agents. In an effort to identify structure-directing agents with selectivity towards novel selenides, we have explored the use of organic superbases, and in particular, of two distinctly different bicyclic amines, 1,8-diazabicyclo[5.4.0]undec-7-ene (DBU) and 1,4-diazabicyclo[2.2.2]octane (DABCO). This work, which is described here, has resulted in the synthesis and characterisation of five new indium selenides (1–5) of varying dimensionality, which exhibit a number of novel structural features, including the incorporation of (Se2)6 anions and of highly unusual [Na(DABCO)2]+ units.

Experimental

Materials and methods

Syntheses were carried out using 23 mL Teflon lined stainless steel autoclaves. The reagents were added individually to the Teflon liner and mixed for approximately 10 minutes, prior to sealing the reaction vessel. The sealed autoclaves were then heated to a set temperature for a defined period of time. After cooling to room temperature, the products were collected by vacuum filtration and washed with deionised water, methylated spirits and acetone.

Synthesis of \([C_6H_{14}N_2][In_2Se_{8.00}(Se_2)_{0.91}]\) (1). A mixture of In (0.1676 g, 1.46 mmol, 99.99%, Aldrich), Se powder (0.2369 g, 3 mmol, 99.99%, Alfa Aesar), DBU (0.3 mL, 2.01 mmol, 98%), imidazole (0.2042 g, 3 mmol, ≥99%) and acetonitrile (3 mL, Fisher) was placed in a sealed autoclave. The vessel was heated to 200 °C for 10 days. The product contained a small amount of a black powder and large yellow crystals. The yellow crystals were identified as 1 by single crystal X-ray diffraction. Elemental analysis: found: C = 17.55%, H = 2.69%, N = 4.38%; calc.: C = 17.84%, H = 2.83%, N = 4.62%.

Synthesis of \([C_9H_{17}N_2][In_5Se_{8.74}(Se_2)_{0.26}]\) (2). This compound was synthesised using InCl3 (0.3229 g, 1.46 mmol, 98%, Aldrich), Se powder (0.2369 g, 3 mmol, 99.99%, Alfa Aesar) and DBU (1 mL, 6.7 mmol, 98%, Aldrich). The sealed vessel was heated at 200 °C for 10 days. The collected product consisted of yellow rod-like crystals of 2. Elemental analysis: found: C = 18.39%, H = 2.95%, N = 4.67%; calc.: C = 18.37%, H = 2.91%, N = 4.76%.

Synthesis of \([C_9H_{17}N_2][C_6H_{14}N_2][In_10Se_{15}(Se_2)_3]\) (3). A mixture of In (0.1676 g, 1.46 mmol, 99.99%, Aldrich), Se powder (0.2369 g, 3 mmol, 99.99%, Alfa Aesar), DABCO (0.6731 g, 6 mmol, ≥99%, Aldrich) and deionised water (3 mL), heated to 200 °C for 14 days. The resulting product contained numerous yellow plates of 3, and a small amount of red needle-like crystals identified by single-crystal X-ray diffraction as the previously reported \([NH_4]^+[In_2Se_{20}]\). The presence of ammonium and ethylenediamine cations in the crystal structure of 5 suggested that DABCO decomposed under solvothermal conditions. A similar decomposition of DABCO has been reported in the hydrothermal preparation of uranium fluorides. Attempts were made to prepare 5 from the products of the decomposition of DABCO. For this purpose, ethylenediamine dihydrochloride (3 mmol, 99%, Aldrich) and ammonium chloride (3 mmol, 99.9%, Aldrich) were used, together with In powder (1.46 mmol), Se (3 mmol) and deionised water (3 mL) at 200 °C for 10 days. This reaction gave a brownish powder where 5 was identified by powder X-ray diffraction, but no single crystals were observed. Elemental analysis for 5: found: C = 1.22%, H = 0.88%, N = 2.29%; calc.: C = 1.31%, H = 0.77%, N = 2.29%.

Single-crystal diffraction

Single-crystal X-ray diffraction data for crystals of 1–4 were collected at 100 K, whilst data for 5 were collected at 293 K. For 3, the crystal used for data collection was particularly small, despite unsuccessful attempts to optimize the reaction conditions to grow larger crystals. The structures of 1–4 were solved using the charge-flipping method, Superflip,14 while that of 5 was solved using SHELX,15 which located the In and Se atoms. Organic molecules were generally found in the difference Fourier maps. Subsequent Fourier calculations and least squares refinements on F were carried out using the CRYSTALS program16 for all structures. Hydrogens were not placed on the ammonium cation in 5 as they are likely to be disordered. Selected crystallographic information for these compounds is shown in Table 1. Data have been deposited with the Cambridge Crystallographic Data Centre as CCDC 1023692–1023696.

Characterisation

Powder X-ray diffraction data were collected using a Bruker D8 Advance powder diffractometer, operating with germanium monochromated CuKα1 radiation (λ = 1.5406 Å) fitted with a Bruker LynxEye linear detector. Data were collected, on a
ground portion of the reaction product, over the angular range 
$5 \leq 2\theta \leq 85$ with a step size of 0.009°. Lattice parameters 
were determined using Topas.$^{17}$ Analysis of powder X-ray diffrac-
tion data (ESI) indicates that $1–5$ are the main products 
obtained. Elemental analysis on handpicked crystals was 
carried out on an Exeter CE-440 elemental analyser. Thermo-
gravimetric analysis was performed using a DuPont Instru-
mements 951 thermal analyser. Approximately 10 mg of finely 
ground crystals were heated under a flow of O$_2$ over the tempe-
trature range $30 \leq T/\text{°C} \leq 1000$. Diffuse reflectance measure-
ments were performed using a Perkin Elmer Lambda 35 UV-vis 
spectrometer. BaSO$_4$ powder was used as a reference (100% 
reflectance) and absorption data were calculated from 
the Kubelka–Munk function.$^{18}$ Infrared measurements were 
carried out using a Perkin Elmer Spectrum 100 ATR 
spectrometer.

### Results

#### Crystal structures of 1 and 2

Although 1 crystallises in the monoclinic space group $P2_1/c$, 
whilst 2 is triclinic ($P1$), their crystal structures are closely 
related, and consist of anionic ribbons separated by proto-
nated DBU cations. The ribbons found in 1 and 2 have the 
compositions $[\text{In}_5\text{Se}_{8.07}\text{(Se}_2)_{0.93}]^{3–}$ and $[\text{In}_5\text{Se}_{8.74}\text{(Se}_2)_{0.26}]^{3–}$ 
respectively, and differ only in the content and arrangement of 
disordered ($\text{Se}_2)^{2–}$ units. The simplest fragment of the anionic 
ribbon in 1 is shown in Fig. 1(a). This repeating unit contains 
five crystallographically-independent tetrahedra linked by their 
edges and corners. In this unit, three tetrahedra containing 
In(1), In(4) and In(5) are linked by their corners forming a six-
membered ring. The In(5)Se$_4$ tetrahedron is linked by an edge 
to an In(3)Se$_4$ tetrahedron, forming a four-membered ring. 
The linkage between In(4)Se$_4$ and In(2)Se$_4$ is more complex,

due to the coexistence of disordered ($\text{Se}_2)^{2–}$ and $\text{Se}^{2–}$ moieties. 
Depending on whether a $\text{Se}^{2–}$ or a ($\text{Se}_2)^{2–}$ unit is present, the 
linkage between In(4)Se$_4$ and In(2)Se$_4$ creates a four- or a five-
membered ring. In 1, the perselenide linkage shown in Fig. 1(a) 
is dominant, with the fractional occupancy of the ($\text{Se}_2)^{2–}$ unit 
being 92.66(18)%, whilst the occupancy of the corresponding

#### Table 1  Crystallographic data for compounds 1–5

<table>
<thead>
<tr>
<th>Compound</th>
<th>1</th>
<th>2</th>
<th>3</th>
<th>4</th>
<th>5</th>
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<tr>
<td>Empirical formula</td>
<td>$\text{C}<em>{27}\text{H}</em>{51}\text{In}_5\text{N}<em>6\text{Se}</em>{9.93}$</td>
<td>$\text{C}<em>{27}\text{H}</em>{51}\text{In}_5\text{N}<em>6\text{Se}</em>{9.26}$</td>
<td>$\text{C}<em>{42}\text{H}</em>{84}\text{In}<em>{10}\text{N}</em>{14}\text{Se}_{21}$</td>
<td>$\text{C}<em>{18}\text{H}</em>{38}\text{In}_5\text{N}<em>6\text{NaSe}</em>{9}$</td>
<td>$\text{C}<em>{2}\text{H}</em>{10}\text{In}_7\text{N}<em>3\text{Se}</em>{12}$</td>
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<tr>
<td>$M_r$</td>
<td>1817.65</td>
<td>1764.67</td>
<td>3603.59</td>
<td>1646.27</td>
<td>1837.38</td>
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<tr>
<td>Crystal habit</td>
<td>Yellow block</td>
<td>Yellow needle</td>
<td>Red block</td>
<td>Red octahedron</td>
<td>Yellow block</td>
</tr>
<tr>
<td>Dimensions/mm$^3$</td>
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<td>$0.04 \times 0.05 \times 0.08$</td>
<td>$0.04 \times 0.04 \times 0.06$</td>
<td>$0.05 \times 0.05 \times 0.09$</td>
<td>$0.05 \times 0.05 \times 0.08$</td>
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<tr>
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<td>$P1$</td>
<td>$P2_1/c$</td>
<td>$C2/c$</td>
<td>$P2_1$</td>
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<td>21.7994(15)</td>
<td>20.189(2)</td>
<td>13.4581(4)</td>
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<td>$b/\text{Å}$</td>
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<td>12.9893(9)</td>
<td>10.490(3)</td>
<td>12.1666(16)</td>
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<td>$c/\text{Å}$</td>
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<td>18.1462(12)</td>
<td>105.196(3)</td>
<td>98.459(6)</td>
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<td>$\alpha/°$</td>
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<td>90</td>
<td>90</td>
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<td>90</td>
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<td>$\beta/°$</td>
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<td>100.004(3)</td>
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<td>99.459(6)</td>
<td>90</td>
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<tr>
<td>$\gamma/°$</td>
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<td>102.004(3)</td>
<td>224.58(3)</td>
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<td>90</td>
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<td>$V/\text{Å}^3$</td>
<td>4633.4(10)</td>
<td>2245.8(3)</td>
<td>8217.4(10)</td>
<td>3674.4(9)</td>
<td>1461.32(7)</td>
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<td>4</td>
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<td>2</td>
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<td>$\rho_{cal}/\text{g cm}^{-1}$</td>
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<td>2.609</td>
<td>2.903</td>
<td>2.976</td>
<td>4.153</td>
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<tr>
<td>Flack</td>
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<td>—</td>
<td>—</td>
<td>0.079(12)</td>
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<tr>
<td>Number of parameters</td>
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<td>261</td>
<td>504</td>
<td>178</td>
<td></td>
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<tr>
<td>$R_{\text{me}}$</td>
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<td>0.093</td>
<td>0.048</td>
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<tr>
<td>$R(I &gt; 3.0\sigma(I))$</td>
<td>0.0389</td>
<td>0.0522</td>
<td>0.0690</td>
<td>0.0270</td>
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<tr>
<td>$R_w$</td>
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<td>0.0572</td>
<td>0.0721</td>
<td>0.0327</td>
<td>0.0344</td>
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</table>

Fig. 1 View of the building block for the $[\text{In}_5\text{Se}_{8.07}\text{(Se}_2)_{0.93}]^{3–}$ ribbons in 
(a) compound 1, and (b) compound 2. For clarity, the disordered $\text{Se}^{2–}$ in 
1, and the disordered ($\text{Se}_2)^{2–}$ in 2, have been omitted. Key: indium, large 
pink circles; selenium, large yellow circles.
ties along the ribbon. In the predominant four-membered rings. Key as for Fig. 1.

InSe₄ tetrahedra are linked via edge-sharing linkages. This creates a number of six-[In₅Se₈], five-[In₄Se₇] and four-[In₃Se₆] membred rings, which were also present in 1 and 2, as well as [In₃Se₄] and [In₅Se₆] rings, not found in 1 and 2. The Se–Se distances in the perselenide (Se₂)²⁻ units present in this material range from 2.411(2) to 2.438(2) Å, which are slightly longer than those reported for (Se₂)²⁻ units in other indium selenides, but within the range of Se–Se distances observed in polyselenides.

The [In₁₀Se₁₅(Se₂)₃]⁹⁺ ribbons in 3 can be described as consisting of two identical one-dimensional chains of In₅Se₄ tetrahedra, related by a rotation of 180°, and fused to form a ribbon (Fig. 4). This ribbon exhibits twists where the tetrahedra are connected via edge-sharing linkages (Fig. 5). In the crystal structure of 3, the ribbons are aligned parallel to the [001] direction and interspaced by DABCO moieties, which surround the ribbons. At each twist in the ribbon, the DABCO moiety is rotated by ∼90° with respect to the other DABCO molecules. This may be related to the presence of hydrogen-
bonding interactions between the amine located at the twist and the ribbon, given the N···Se distance of 3.27(3) Å, which is similar in magnitude to previously reported hydrogen-bonding interactions in selenides.\textsuperscript{6,12a,20} Charge balance requires protonation of some of the organic molecules, but the structural study is unable to establish directly the degree of protonation of the amines. Given the significantly shorter N···Se distance (3.27(3) Å) found for the DABCO at the twist when compared to the remaining crystallographically independent amines (>4.1 Å), the former is likely to be protonated. There are also short N···N distances (ca. 2.7 Å) between neighbouring DABCO moieties, consistent with N···H···N bonding interactions.

**Crystal structure of 4**

Compound 4 is the first example of incorporation of a sodium complex into a solvothermally-prepared indium selenide. The crystal structure of this material contains one-dimensional ribbons, linked into a three-dimensional structure by \([\text{NaSe}_3(\text{DABCO})_2]^{5-}\) complexes. In 4, edge and corner linkages between \(\text{InSe}_4\) tetrahedra result in the formation of a one-dimensional ribbon, with stoichiometry \([\text{In}_3\text{Se}_9]^{5-}\). This ribbon is closely related to those found in 1 and 2, although in this case, no perselenide \((\text{Se}_2)^{2-}\) anions are present (Fig. 2(b)).

The ribbons are oriented along the [101] direction and linked into a three-dimensional network (Fig. 6(a)) by trigonal bipyramidal complexes of \([\text{NaSe}_3(\text{DABCO})_2]^{5-}\). In this complex, sodium is bonded to three selenium atoms from three neighbouring ribbons, in the equatorial positions and two nitrogen atoms from two DABCO molecules, in the axial positions (Fig. 6(b)). The Na–Se distances are 2.9681(13) and 2.973(3) Å, comparable to those in \([\text{NaSe}_3(\text{DABCO})_2]^{2-}\) and the Na–N distance is 2.516(3) Å, longer than the Na–N bonds in \([\text{NaSe}_3(\text{DABCO})_2]^{-}\). This might be a consequence of steric interactions between the ligands. This sodium complex is extremely unusual. The closest precedent is found in a small number of thiolates in which sodium is coordinated to N and S atoms with a coordination number of 5, but their geometry is better described as that of a distorted square planar pyramid.\textsuperscript{23} To the best of our knowledge, examples containing selenium are not known.

To achieve charge balance, one diprotonated DABCO cation per formula unit is incorporated into the structure of 4. There are relatively short N···N distances between the DABCO cations and the \([\text{NaSe}_3(\text{DABCO})_2]^{5-}\) complexes, of 2.705(5) Å, which imply the presence of hydrogen-bonding interactions. Although 4 exhibits a three-dimensional structure, when van der Waals’ radii are taken into account there is no available porosity or free channels. This material hence consists of a dense framework.

**Crystal structure of 5**

Compound 5 is a rare example of a three-dimensional indium selenide. The asymmetric unit of 5 (ESI) contains seven crystallographically-independent indium atoms, all of which are tetrahedrally coordinated to selenium. The simplest secondary building block in 5 is shown in Fig. 7(a). In this unit, tetrahedra containing In(2), In(4) and In(5) are corner-linked forming a chain. This is linked to another chain of corner-linked tetrahedra containing In(3), In(6) and In(7). The In(1)Se$_4$ tetrahedron caps this double chain by corner-linking to In(2), In(3) and In(7)Se$_4$ tetrahedra. These units of seven tetrahedra are linked into two-dimensional buckled layers (Fig. 7(b)) by the terminal Se vertices. Rotation by 90° of the layers depicted in Fig. 7(b), followed by their linkage through the corner of the capping tetrahedron, \((\text{In}(1)\text{Se}_4))\, results in the formation a three-dimensional framework, illustrated in Fig. 7(c).

For this anionic framework, the solvent accessible void space, calculated using the SOLV function in PLATON,\textsuperscript{24} is only 18.9% of the unit cell volume. This void space consists of small cavities where protonated ethylenediamine and ammonium cations reside. There is a number of N···Se distances indicative of hydrogen-bonding interactions between the framework and the organic moieties. Each of the \(\text{NH}_3^+\) groups of the en$_2$H$_2$\textsuperscript{2+} cation are involved in three H-bonding interactions with distances in the range 3.35(2) to 3.57(3) Å, with the \(\text{NH}_4^+\) cation involved in four N···H···Se interactions of 3.34(1) to 3.59(1) Å which are comparable to interactions in previously reported indium and gallium selenides.\textsuperscript{6,12a,20}
X-ray crystallography. Bands observed for

\[ \nu(C=N) \] and \[ \delta(C-H) \] respectively. Bands at lower wavenumbers (1320 to 2900 cm\(^{-1}\)) are consistent with deformation and bends of C–N

\[ \nu(N-H) \] and C–C bonds. Bands in 3 and 4 are consistent with the presence of protonated and unprotonated DABCO, as established by comparison with the assignments reported by Guzonas and Irish.\(^{25}\) FT-IR data for 5 are consistent with the presence of the \([\text{NH}_4]^+\) cation within the crystal structure. The band at ~3000 cm\(^{-1}\) can be assigned to the N–H symmetric stretching vibration, with the signal at 1600 cm\(^{-1}\) corresponding to N–H deformation of the \([\text{NH}_4]^+\) cations. The signal 1400 cm\(^{-1}\) can be assigned to an N–H bending mode.\(^{26}\) Other peaks in the spectra are consistent with the presence of protonated ethylenediamine within the crystal structure (~1524 cm\(^{-1}\) – N–H

def, 1478 cm\(^{-1}\)).

**Optical absorption properties**

UV-Vis diffuse reflectance measurements (Fig. 8) were carried out to estimate the optical band gap of compounds 1–5. The band gap values, which are given in Table 2, are consistent with the colour of the crystals (Fig. 8). For antimony sulphides, the band gap variation has been related to the density of metal

<table>
<thead>
<tr>
<th>Compound</th>
<th>Band gap/eV</th>
<th>In atoms per 1000 (Å(^3))</th>
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</thead>
<tbody>
<tr>
<td>1</td>
<td>2.49(3)</td>
<td>4.3</td>
</tr>
<tr>
<td>2</td>
<td>2.83(4)</td>
<td>4.5</td>
</tr>
<tr>
<td>3</td>
<td>1.88(3)</td>
<td>4.9</td>
</tr>
<tr>
<td>4</td>
<td>1.79(1)</td>
<td>5.4</td>
</tr>
<tr>
<td>5</td>
<td>2.44(2)</td>
<td>9.6</td>
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</tbody>
</table>

**Characterisation**

Powder X-ray diffraction data (ESI) collected on the bulk products of the reactions producing 1–5 indicate that 1–5 are the main products of the reactions, and the lattice parameters (ESI) determined for 1–5 using powder X-ray diffraction data are in good agreement with those determined by single-crystal diffraction.

Thermogravimetric analysis (ESI) was carried out on handpicked crystals of 1–5 under an oxidizing atmosphere. Compounds were found to be stable up to 180–250 °C. The weight loss occurs as a multistage process, with the overall weight change corresponding to the decomposition of the starting material into In\(_2\)O\(_3\). For 1, 3 and 5, a small increase in weight is observed between 300 and 400 °C, suggesting that a selenate, selenite or oxyselenide may form in the initial stages of the decomposition process. The found weight losses, together with those calculated for the removal of the amine are as follows: 1 found: 61.2%; calc.: 61.8%; 2 found: 62.2%; calc.: 60.7%; 3 found: 59.8%; calc.: 61.5%; 4 found: 57.2%; calc.: 57.8%; 5 found: 48.3%; calc.: 46.9%.

FT-IR data (ESI) collected for 1 to 5 show bands consistent with the organic species identified within the structure by X-ray crystallography. Bands observed for 1 and 2 are consistent with the presence of DBUH\(^+\). For instance, the strong bands at 1685 to 1580 cm\(^{-1}\) correspond to the C–N stretch, that at ~2900 cm\(^{-1}\) can be assigned to C–H stretches, whilst the bands at 1632 and 1443 cm\(^{-1}\) are characteristic of \(\nu(C=\text{N})\) and \(\delta(C–\text{H})\) respectively. Bands at lower wavenumbers (1320 to 1000 cm\(^{-1}\)) are consistent with deformation and bends of C–N

Fig. 7 Polyhedral representation of (a) the secondary building block in 5; (b) the zig-zag layer in 5, with the secondary building block circled, and (c) view of the crystal structure of 5 along the [100] direction. Key: In(1)Se\(_4\), blue; In(2)Se\(_4\), green; In(3)Se\(_4\), teal; In(4)Se\(_4\), purple; In(5)Se\(_4\), turquoise; In(6)Se\(_4\), pink; In(7)Se\(_4\), red; carbon, small green circles; nitrogen, small blue circles. Hydrogen atoms have been omitted for clarity.

Fig. 8 UV-Vis diffuse reflectance spectra for (a) compounds 1–3 and (b) 4 and 5 (left). Photographs of the crystals of 1–5 (right).
centres within the crystal structures. This correlation has been attributed to the weakening of the interactions between anion and cation orbitals with decreasing density, which has an effect on the energy levels at the top of the valence band. As shown in Table 2, this simple relationship between band gap and density of metal centres (given as number of In atoms per 1000 Å³) does not hold here. We have previously observed, when comparing the related chains [GaSe₂]⁻ and [Ga₂Se₆(Se₂)]⁻, a marked reduction of the band gap when (Se₃)²⁻ anions are present. Similarly, the band gap of [DBUH][Hg₂Sn₂Se₆(Se₂)] is lower than that of the closely related [DBUH][Hg₂Sn₂Se₇]. This suggests that the perselenide anion contributes to energy levels close to the band gap, and hence may explain the larger band gap found for 5, when compared to those of the materials containing perselenides (1–3). The band gap variation between 4 and 5 does not follow either the expected trend of a smaller band gap with increasing framework density. This is perhaps not surprising, given that in these materials the connectivity between tetrahedra is very different and they also differ in the presence of Na.

## Discussion

### Synthetic considerations

The preparation, under mild solvothermal conditions, of compounds containing sulfur–sulfur bonds occurs only rarely. Despite the large number of main-group sulfides prepared by solvothermal synthesis, examples containing S–S bonds remain scarce. By contrast, recent literature provides growing evidence of the relative ease of formation of compounds containing Se–Se bonds under solvothermal conditions. For example, (Se₃)²⁻ anions have been found in a variety of chains, including [M₂Se₅(Se₂)]⁻ (M = Ga, In), [Ga₂Se₆(Se₃)(Se₂)]⁻ and [Ga₂Se₆(Se₂)]⁻, and in heterometallic ribbons such as [DBUH][Hg₂Sn₂Se₆(Se₂)] and [DBNH][Cd₆Sn₄Se₁₁(Se₂)] where DBN = 1,5-diazabicyclo[4.3.0]non-5-ene, as well as the compounds 1–3 as reported here. This contrasting behaviour may be related to the greater tendency shown by the heavier chalcogens to form chalcogen-rich phases, containing chalcogen–chalcogen bonds. Examination of the synthetic conditions suggests that the preparation of phases containing (Se₃)²⁻ occurs more readily when more basic amines are used as solvents and/or structure-directing agents. For instance, we have observed that reactions using substituted pyridines as solvents, with pKₐ values of ~5 to 7, have resulted in materials that do not contain perselenides, while reactions using primary amines (e.g. 1,4-diaminobutane, pKₐ ~ 10.8) or amidines, which are strongly basic, have produced perselenides. Of the amines used here, DBU (pKₐ ~ 24.3) is more strongly basic than DABCO (pKₐ ~ 8.9), which has a pKₐ comparable to that of some alkyl amines. In both cases, materials containing (Se₃)²⁻ have been obtained. Potential/pH diagrams for selenium in aqueous solutions at ambient conditions indicate that polyselelenides are predominant species at high values of pH. A similar trend seems to occur under solvothermal conditions. Further experimental data on selenides synthesised using different structure-directing agents will be useful to confirm the relationship between pKₐ and the presence of (Se₃)²⁻ anions.

With DBU as a structure-directing agent, compounds 1 and 2, which differ in their (Se₃)²⁻ content, were obtained by varying the indium source and solvent, as well as adding imidazole, which is not incorporated in the final crystal structure. This suggests that other reaction parameters can influence the amount of (Se₃)²⁻ anions incorporated into the crystal structure. It has been previously shown that the one-dimensional [DBUH][Hg₂Sn₂Se₆(Se₂)] transforms into [DBUH][Hg₂Sn₂Se₇] when increasing the reaction time. Attempts to control the (Se₂)²⁻ content in the [In₃Se₆(Se₂)]⁻ ribbons by varying the time were carried out, but these have not been successful so far. While DBU was found to be stable at the reaction temperatures investigated (115–200 °C), DABCO only remained intact when low reaction temperatures were used. It was found that reactions using DABCO afforded a variety of structures of differing dimensionality, including the previously reported frameworks [In₁₃Se₅K]⁻ and [In₁₀Se₁₈]⁻, which were obtained at 170 and 185 °C respectively (see ESI†). A summary of the compounds obtained using DABCO is given in Scheme 1. As the temperature of the reaction is raised, an increase in the dimensionality of the products, together with an increase in the density of metal centres, is observed. At 140 °C, the materials prepared, 3 and 4, contain one-dimensional ribbons, while at intermediate temperatures, the microporous frameworks [In₃₃Se₅₆]⁻ and [In₁₀₅Se₁₈]⁻, both of which have a porosity of ca. 50% of the available unit cell volume, are obtained. Compound 5, which was prepared at the highest temperature investigated, 200 °C, consists of a dense three-dimensional framework with no accessible porosity. Similar findings have been reported for the solvothermal synthesis of cobalt succinates and for hybrid vanadium oxyfluorides, where increased condensation and density was found as reaction temperatures increased. Although there are no systematic studies on the effect of temperature on the solvothermal synthesis of indium selenides, similar results have been found for other metal chalcogenides. In particular, the solvothermal synthesis, in the presence of ethylenediamine, of antimony sulfides containing transition-metal complexes has been extensively explored. In this system, it has been shown that a change of only 25 °C can result in the condensation of isolated [Sb₂S₅]²⁻ anions into [Sb₂S₇]⁻ chains, although changes in reagent concentration whilst maintaining a constant temperature can also result in markedly different products. Similarly, for tin chalcogenides, marked changes in the structure dimensionality have been achieved at room
temperature by varying the reactants and solvent concentrations.\textsuperscript{29}

**Structural chemistry**

The anionic ribbons found in 1–4 are the first examples of ribbons in indium selenides. Two distinct ribbons, both of which constitute new structural types, have been identified in this work. Compounds 1, 2 and 4 contain anionic ribbons with the general formula \([\text{In}_5\text{Se}_8]^\text{+}\), while 3 contains a \([\text{In}_3\text{Se}_6\text{Se}_3]^\text{3–}\) ribbon. In both cases, corner- and edge-sharing linkages of \(\text{InSe}_4\) tetrahedra coexist in the ribbon, and this can lead to a long repeating sequence along the ribbon, as found in 3, where the distances between consecutive twists is ca. 36 Å. In the case of the \([\text{In}_2\text{Se}_{10}\text{Se}_2]^\text{3–}\) ribbons, partial substitution of \(\text{Se}^\text{2–}\) with \(\text{Se}_2\text{Se}^\text{2–}\) occurs in the \([\text{In}_3\text{Se}_6]\) rings in 1 and 2, whilst 4 contains no perselenide. Although the complete replacement of \(\text{Se}^\text{2–}\) with a perselenide has been observed in simple indium selenide chains,\textsuperscript{12} partial substitution had not been previously encountered. The existence of ribbons with \(x\) values of 0, 0.26 and 0.93 suggests that the synthesis of analogous ribbons with different perselenide contents should be possible, and this could be exploited to control the properties (e.g. optical band gap) of these materials.

Although there are no prior reports of indium selenide ribbons in the literature, anionic ribbons, with entirely different structural motifs to the compounds described here, have been found in the tin selenides \([\text{DBUH}]\text{[Hg}_2\text{Sn}_2\text{Se}_6\text{Se}_2]\text{[DBUH]}\text{[Hg}_2\text{Sn}_2\text{Se}_7]\text{[DBUH]}\) as well as in the tellurides \([\text{Na}(\text{DABCO})_2]\text{[Hg}_2\text{Sn}_2\text{Se}_7]\text{[DABCO]}\text{[Hg}_2\text{Sn}_2\text{Te}_7]\text{[DABCO]}\) where \(\text{DABCO}\) = 1,4-diazabicyclo[2.2.2]octane and \(\text{DAP}\) = 1,3-diaminopropane.\textsuperscript{30}

The mercury tin selenides differ from the ribbons reported here in the absence of edge linkages between tetrahedra, while in the indium tellurides, a combination of corner- and edge-sharing also occur. Whilst \([\text{HTMDP}]_2[\text{In}_2\text{Te}_3]\) does not contain \((\text{Te}_2)^\text{2–}\) units, the structure of \([\text{HDAP}]_8[\text{In}_{12}\text{Te}_{23}]\) consists of helical double chains of \(\text{InTe}_4\) tetrahedra connected into a 3-dimensional network by \([\text{In}_3\text{Te}_6]\) rings which contain a \((\text{Te}_2)^\text{3–}\) anion.\textsuperscript{30}

In compound 4, the indium selenide ribbons are linked into a 3-dimensional network by highly unusual \([\text{Na}(\text{DABCO})_2]\text{[Hg}_2\text{Sn}_3\text{Se}_6]\text{[DABCO]}\text{[Hg}_2\text{Sn}_3\text{Te}_6]\text{[DABCO]}\) units. While \(\text{Na}^+\) ions can act as counterions, as exemplified by the indium selenide framework ICF-21,\textsuperscript{2} the incorporation of an alkali-metal complex as a bridging unit into a solvothermally-prepared chalcogenide is unprecedented. Transition-metal complexes have been shown to act as counterions, pendants groups or linkers.\textsuperscript{41} By contrast, there are very few known examples of chalcogenides containing complexes of group 1 and 2 elements. These appear to be limited to the \([\text{Mg}(\text{en})_3]\text{[Sb}_2\text{S}_7]\) and \([\text{Mg}(\text{en})_3]\text{[In}_2\text{S}_4]\)\textsuperscript{42} and the \([\text{Na}_3(\text{H}_2\text{O})_9]\text{[S}_2\text{S}_7]\) clusters, which fill the cavities in the 3-dimensional framework of \([\text{Na}_3(\text{Mg})_3(\text{S})_3\text{[S]}_2\text{S}_7]\text{[H}_2\text{O}]\text{[M = Ga, In]}\).\textsuperscript{13}

Compound 5 also exhibits a three-dimensional structure, which arises from corner linkage of \(\text{InSe}_4\) tetrahedra. Only a small number of indium selenide frameworks are known,\textsuperscript{3,7,9,10} and 5 differs from previously reported examples in its lack of channels or accessible porosity. This material, which is obtained at temperatures at which \(\text{DABCO}\) decomposes, consists of a highly dense framework, containing only small cavities.

**Conclusions**

We have demonstrated that bicyclic amines can be used for the preparation of complex indium selenides, containing perselenide \((\text{Se}_2)^\text{2–}\) anions, and identified amine \(pK_a\) as a significant parameter for the targeted synthesis of perselenides. In addition, the thermal stability of the amine under solvothermal conditions is important in order to produce new structural types. We have also shown that alkali-metal complexes can act as linkages in indium selenide frameworks, and identified a range of entirely new structural building blocks in these materials. These findings indicate that a wide range of complex indium selenides, containing \((\text{Se}_2)^\text{2–}\) anions and/or alkali-metal complexes, should be accessible through mild solvothermal reactions.

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**Notes and references**
